

REPORT DOCUMENTATION PAGE

AFRL-SR-BL-TR-98-

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1. AGENCY USE ONLY (Leave blank)		2. REPORT DATE 9/98		3. REPORT TYPE AND DATES COVERED Final Technical Report 6/95-5/98	
4. TITLE AND SUBTITLE GaAs-GaInP Superlattices for Intersubband Photodetection				5. FUNDING NUMBERS F4960-95-1-0422	
6. AUTHOR(S) Dr. Manijeh Razeghi					
7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) Northwestern University Center for Quantum Devices 2225 N. Campus Drive, MLSB Room 4051 Evanston, IL 60208-3118				8. PERFORMING ORGANIZATION REPORT NUMBER	
9. SPONSORING/MONITORING AGENCY NAME(S) AND ADDRESS(ES) U.S. Air Force Office of Scientific Research/NE 110 Duncan Avenue Suite B115 Bolling Air Force Base, DC 20332-6448				10. SPONSORING/MONITORING AGENCY REPORT NUMBER	
11. SUPPLEMENTARY NOTES					
12a. DISTRIBUTION/AVAILABILITY STATEMENT				12b. DISTRIBUTION CODE	
13. ABSTRACT (Maximum 200 words) We have demonstrated the first QWIP detectors using the quaternary InGaAlAs/InP materials system. We have observed excellent responsivity in the InGaAlAs/InP system compared to GaAs/AlGaAs QWIPs. By increasing the bandgap from ternary InGaAs to quaternary InGaAlAs we have shifted the responsivity out to longer wavelengths resulting in cutoff wavelengths of 13.3 and 19.4 μm s, respectively for AlAs mole fractions of 0.1 and 0.15. We have also demonstrated lattice-matched mid-wavelength infrared detectors using InGaAs/InAlAs quantum wells. By combining both types of devices, we have produced the first lattice-matched dual band mid-wavelength and long-wavelength QWIP detectors on InP substrate. We have also demonstrated long-wavelength QWIPs ($\lambda > 12 \mu\text{m}$) in GaAs-GaInP structures, and theoretically calculated the performance of these devices.					
14. SUBJECT TERMS				15. NUMBER OF PAGES	
16. PRICE CODE					
17. SECURITY CLASSIFICATION OF REPORT		18. SECURITY CLASSIFICATION OF THIS PAGE		19. SECURITY CLASSIFICATION OF ABSTRACT	
				20. LIMITATION OF ABSTRACT	



“GaAs-GaInP Superlattices for Intersubband Photodetection”

AFOSR Contract # F4960-95-1-0422 / NU #0650-350-S402

Project Manager: Dr. Gerald Witt

FINAL TECHNICAL REPORT (AASERT)

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Outline

1. Introduction
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Introduction

Background

Infrared detection and thermal imaging are topics of strong interest for military programs, both for night vision and surveillance and also for weapon aiming. Alongside this is the rapid growth in the civil and commercial fields of applications of this technology which stretch from crime prevention to crop disease detection. For both military and civil applications, large staring arrays of infrared detectors provide the best spatial and temperature resolution images. This is reflected in the numerous theoretical and experimental studies of materials for use as detectors in infrared arrays that have steadily advanced the performance of these imaging systems.

The most challenging detectors to manufacture are those which are sensitive to infrared radiation at wavelengths greater than eight microns. The need for these long-wavelength infrared (LWIR) detectors derives from the fact that the infrared energy density radiated from objects at ambient temperatures reaches a maximum in the 8-14 micron region.

At present, advances in microelectronic fabrication and semiconductor crystal growth technology has allowed new physical phenomenon based on quantum mechanics to be used for infrared detection. Very thin semiconductor layers called quantum wells are used to construct quantum well infrared photodetectors (QWIPs) operating in the LWIR. Although at present they require lower operating temperatures to achieve the same sensitivity as other infrared sensors, QWIPs have several advantages including better uniformity, better large array manufacturability, and lower manufacturing costs.

The efforts of this AASERT were to provide a better experimental data base for a variety of new QWIP materials systems based on quantum wells and barriers made from $\text{Ga}_x\text{In}_{1-x}\text{As}_y\text{P}_{1-y}$ materials. Currently, there is limited knowledge about QWIPs based on materials other than $\text{GaAs}/\text{Al}_x\text{Ga}_{1-x}\text{As}$. New QWIP designs intended to improve sensitivity, especially designs developed to achieve multi-color, necessitate the use of these new quantum well materials. Thus the results of these studies will have direct technological significance by providing experimental growth, characterization, fabrication guidelines as well as a basic understanding of detector performance for these new materials.

Objectives

General:

- High uniformity, background limited, multi-color detector arrays (focal plane arrays) operating in the infrared (IR) region between 3 and 20 μm .

Specific:

- Explore the use of GaAs and InP-based materials such as GaInP, InGaAs, InAlAs, InGaAsP, and InGaAlAs in quantum well infrared photodetectors (QWIP) designs and demonstrate improved performance by comparison to AlGaAs-GaAs QWIPs over the IR range $3\ \mu\text{m} < \lambda < 20\ \mu\text{m}$.
- Develop very-long wavelength ($\lambda > 20\ \mu\text{m}$) QWIP detectors with better performance than mercury cadmium telluride (HgCdTe) detectors and higher temperature operation than cooled extrinsic detectors.
- Develop multi-color detectors lattice-matched to InP or GaAs with each pixel simultaneously sensitive to light in three IR wavelength bands: MWIR 3-5 μm , LWIR 8-12 μm and VLWIR : 12-20 μm .

Applications

Defense applications: Night vision, camouflage countermeasures, land mine detection, reconnaissance, and space-base imaging.

Civilian applications: Industrial manufacturing inspection, search and rescue and firefighting, medical imaging.

Major Achievements

For InP-based QWIPs:

- Record-high responsivity $> 8 \text{ A/W}$ at $T=77\text{K}$ for InGaAs/InP QWIPs.
- First experimental measurements of bias-dependant gain, quantum well capture probability, and recombination rate for InP-based QWIPs.
- Lower recombination rate results in higher detectivity, which allows higher temperature operation for LWIR detectors.
- 50X higher gain for InGaAs/InP vs. AlGaAs/GaAs indicates improved transport and carrier lifetime are obtained in InP-based materials.
- InGaAs/InAlAs 3-5 μm MWIR QWIPs demonstrate responsivity to 250K and record narrow FWHM of 0.13 μm .
- First demonstration of InGaAlAs/InP 8-20 μm LWIR QWIPs.
- First demonstration of two-color (MWIR and LWIR) QWIP detector lattice-matched to InP substrate.

For GaAs-based QWIPs:

- Fabrication of QWIPs with peak responsivity at very long wavelengths, $\lambda_c \sim 15 \mu\text{m}$.
- Improved performance levels to $D^* = 3 \times 10^8$ at $T=77\text{K}$, quantum efficiency. $\sim 14 \%$.
- Completed first theoretical modeling of GaAs-GaInP QWIPs. I-V curves; drift velocity, mobility carrier lifetime calculated from this model.
- Deliverables: First QWIP focal plane array in GaAs/GaInP demonstrated, and delivered to NASA-Goddard for testing.

Publications supported by this AASERT Grant

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InP-Based QWIP Results

I. Background

For some applications, MQW detectors lattice-matched to InP substrates have several advantages in comparison to LWIR MQW detectors composed of GaAs/Al_xGa_{1-x}As:

- (1) The effective mass of electrons, which governs the drift mobility and tunneling properties, amounts to $0.042 \cdot m_0$ in In_{0.53}Ga_{0.47}As/InP compared to $0.067 \cdot m_0$ GaAs/AlGaAs, where m_0 is the free-electron mass;
- (2) Mid-wavelength infrared (MWIR) 3-5 μm In_{0.53}Ga_{0.47}As/Al_{0.48}In_{0.52}As quantum well detectors are lattice-matched with LWIR In_{0.53}Ga_{0.47}As/InP QWIP detectors, allowing a 2-color MWIR-LWIR lattice-matched detector stack to be grown on InP. This minimum intersubband absorption wavelength that can be achieved in GaAs/AlGaAs QWIPs is 6 μm . [1]

Since the photoresponse spectrum of an In_{0.53}Ga_{0.47}As/InP QWIP has a maximum at approximately 8 μm , [6-9] the use of In_xGa_{1-x}As_yP_{1-y} quantum wells has been studied [2] to allow longer wavelength ($\lambda \sim 12 \mu\text{m}$) detection. Because the InGaAlAs material system covers approximately the same bandgap energies as InGaAsP, it is interesting to compare the hot electron transport and performance of quantum well infrared photodetectors fabricated from these two materials.

The bandgap of InGaAlAs can be engineered between the two boundary ternary alloys, In_{0.53}Ga_{0.47}As (0.76 eV) and In_{0.52}Al_{0.48}As (1.46 eV). The growth of InGaAlAs alloy is relatively easier than InGaAsP due to the following reasons [3]: (1) only one group V element (As) is incorporated, avoiding the problem of As/P ratio control; (2) composition of the layer is controlled by each constituent element's flux intensity; and (3) near-unity sticking coefficients of the three group III elements facilitate reproducibility of composition.

II. Device Design

For this experiment, six structures were grown by gas-source molecular beam epitaxy with arsine and phosphine as group V sources, elemental gallium and indium as group III sources, and elemental silicon as an *n*-type dopant source. The AlAs mole fraction in In_{0.52}(Ga_{1-x}Al_x)_{0.47}As and the quantum well width for the three long-wavelength samples were ($x=0$, 56 Å), ($x=0.1$, 60 Å), and ($x=0.15$, 65 Å) respectively. Each structure consisted of twenty periods of InGaAlAs quantum wells separated by 400 Å InP barriers. Two mid-wavelength samples of In_{0.53}Ga_{0.47}As/In_{0.52}Al_{0.48}As were grown with quantum well widths of 35 and 40 Å. Each structure consisted of twenty periods of In_{0.53}Ga_{0.47}As quantum wells separated by 400 Å In_{0.53}Al_{0.48}As barriers. Lastly, a multi-color structure consisted of 20 periods of In_{0.53}Ga_{0.47}As/InP (well width = 58 Å, barrier width = 500 Å) and 20 periods of In_{0.53}Ga_{0.47}As/In_{0.52}Al_{0.48}As (well width = 35 Å, barrier width = 400 Å). The quantum wells for all devices were doped $n=8 \times 10^{17} \text{ cm}^{-3}$. Top and bottom 0.6 μm layers of $n=1 \times 10^{18} \text{ cm}^{-3}$ In_{0.53}Ga_{0.47}As were grown for ohmic contacts. Arrays of 400 μm^2 mesas were etched in a Plasma-Therm 770 ECR/RIE etch system. A mix of H₂/Ar/Cl₂/CH₄ gases was used. Finally, AuGe/Ni/Au ohmic contacts were evaporated onto the top and bottom contact layers.

III. Long-Wavelength Infrared Detector

The relative responsivity spectra for these three samples were collected using a Mattson Galaxy 3000 FTIR. The responsivity spectrum of the three samples measured at $T=10\text{K}$ are shown in Fig. 1. For a bias of -1 V (mesa top negative), the 50% long wavelength cutoff wavelengths for the three samples were 8.5, 13.3, and 19.4 μm , respectively. The spectral response was unchanged for all three samples in the temperature range from $T=10\text{K}$ up to $T=77\text{K}$.

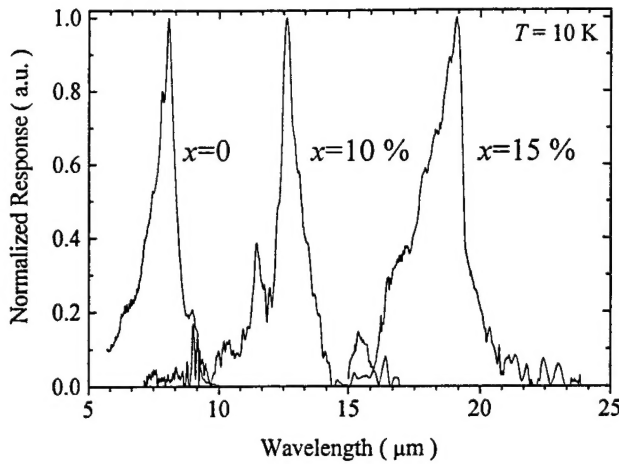


Fig. 1. Normalized spectral response for the three $\text{In}_{0.52}(\text{Ga}_{1-x}\text{Al}_x)_{0.48}\text{As}/\text{InP}$ QWIPs with $x=0$, $x=0.1$, and $x=0.15$ mole fraction of AlAs.

The absolute responsivities were measured by using a calibrated blackbody source. The detectors were back illuminated through a 45° polished facet. The bias dependence of the responsivity was measured for samples A and B at $T=77\text{K}$ and the results are shown in Fig. 3 for both positive and negative bias. The responsivity of sample C was too low to be measured at $T=77\text{K}$. The peak responsivity at -1 V of the $\text{In}_{0.52}\text{Ga}_{0.38}\text{Al}_{0.1}\text{As}/\text{InP}$ QWIP (sample B) was 0.37 A/W . This is comparable (20 % higher) to the $\text{InGaAsP}/\text{InP}$ QWIP ($1.3\text{ }\mu\text{m}$ bandgap, $L_w=63\text{ }\text{\AA}$) reported by Gunapala *et al.* [2] which had a similar $13.2\text{ }\mu\text{m}$ cutoff wavelength. It is worth noting the responsivity for these samples are approximately five times as large as the best responsivity measured for $\text{GaAs}/\text{AlGaAs}$ QWIPs.

The noise current i_n of the $\text{In}_{0.52}\text{Ga}_{0.38}\text{Al}_{0.1}\text{As}/\text{InP}$ and $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{InP}$ QWIPs were measured at $T=77\text{K}$ as a function of bias voltage using a spectrum analyzer and found to be $i_n=23$ – and 0.3 -pA at bias voltages of $V_B=-0.5\text{ V}$. The peak detectivities, D^*_λ can now be calculated from $D^*_\lambda = R\sqrt{A\Delta f}/i_n$ where $A=1.6\times 10^{-4}\text{ cm}^2$ is the device area and $\Delta f=1\text{ Hz}$ is the bandwidth. At an operating bias of $V_B=-0.5\text{ V}$ and $T=77\text{K}$ the measured values for the $\text{In}_{0.52}\text{Ga}_{0.38}\text{Al}_{0.1}\text{As}/\text{InP}$ QWIP are $R_p=0.16\text{ A/W}$, $i_n=23\text{ pA}/\sqrt{\text{Hz}}$ and thus $D^*=3\times 10^8\text{ cm}^2/\sqrt{\text{Hz W}}$. The detectivity of the first sample (GaInAs/InP QWIP) at $T=77\text{K}$ and $V_B=-0.5\text{ V}$ is $D^*=3.5\times 10^{10}\text{ cm}^2/\sqrt{\text{Hz W}}$.

Using the spectral response data obtained for these samples, it is possible to estimate the conduction band offset for the $\text{In}_x\text{Ga}_{1-x-y}\text{Al}_y\text{As}/\text{InP}$ heterojunction. The cutoff wavelength for samples A, B, and C correspond approximately to the energy separation $\Delta E=E_2-E_1$ for the three aluminum compositions, $y=0$ (A), $y=0.1$ (B), and $y=0.15$ (C). The energies of the first and second allowed states for these $\text{InGaAlAs}/\text{InP}$ samples can be calculated for several possible conduction band offsets using the well widths given in part II. The E_g of $\text{In}_{0.52}(\text{Ga}_{1-x}\text{Al}_x)_{0.48}\text{As}$ layers was reported by Fujii *et al.* [4], and changes linearly from 0.75 eV to 1.47 eV with increasing x . The electron effective mass (m_e) of $\text{In}_{0.52}(\text{Ga}_{1-x}\text{Al}_x)_{0.48}\text{As}$ layers was reported by Olego *et al.* [5], and changes linearly from 0.041 to 0.075 with increasing x . A plot of the fit conduction band offset as a function of aluminum fraction is shown in Fig. 3. The

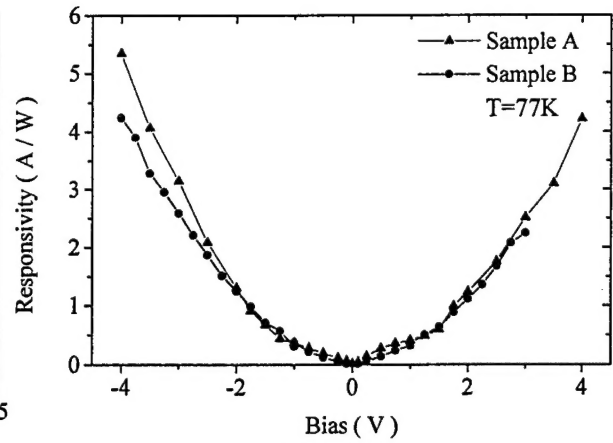


Fig. 2. Bias dependence of peak responsivity measured at $T=77\text{K}$ for $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{InP}$ and $\text{In}_{0.52}\text{Ga}_{0.38}\text{Al}_{0.1}\text{As}/\text{InP}$ QWIPs. No response was observed for the $\text{In}_{0.52}\text{Ga}_{0.33}\text{Al}_{0.15}\text{As}/\text{InP}$ QWIP at $T=77\text{K}$.

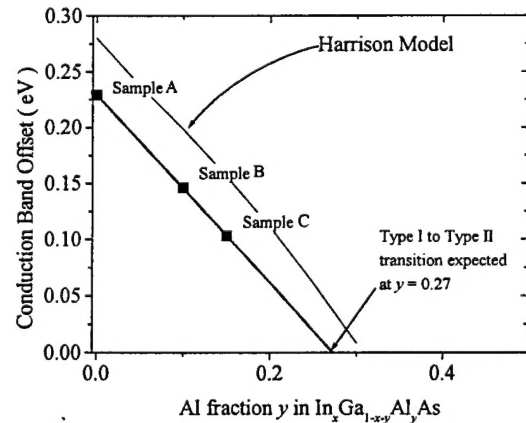


Fig. 3. Conduction band offset values from samples as a function of Al fraction y in $\text{In}_x\text{Ga}_{1-x-y}\text{Al}_y\text{As}$. Also shown is a theoretically derived curve by Ishikawa [10].

data from samples A, B, and C are indicated on the figure. The line is the band offset predicted by the Harrison model as applied by Ishikawa *et al.* [6-7].

From the experimental data for these QWIPs, it would be expected that the transition from type I quantum well to type II staggered quantum well for $\text{In}_{0.53}(\text{Ga}_x\text{Al}_{1-x})_{0.47}\text{As}/\text{InP}$ heterojunctions would occur at $y=0.27$. This is less the value ($y=0.33$) predicted by the experimental model, which is based on an interpolation from binary data for InAs, GaAs, and AlAs, but larger than other recent experimental results for GSMBE-grown material: $y=0.18$ calculated from interband absorption by Kawamura, *et al.* [8], and $y=0.23$ calculated from Shottky diode dark currents by Chua *et al.* [3]. Because intersubband absorption is very sensitive to the conduction band offset, the measurements presented here represent an accurate method for determining this type I-to-type II transition composition.

IV. Mid-Wavelength Infrared Detector

The responsivity spectrum of the two mid-wave samples measured at $T=77\text{K}$ are shown in Fig. 4. For a bias of -1 V (mesa top negative), the 50% long wavelength cutoff wavelengths for the three samples were 4.05 and 4.32 μm , respectively. The absolute responsivity saturated at -4 V bias for both samples, with a peak value of $\sim 35 \text{ mA/W}$. The magnitude of the responsivity did not vary with temperature in the range $T=77\text{K}$ to $T=200\text{K}$, as shown in Fig. 5. The maximum detectivity obtained at $T=77\text{K}$ for these samples was $\sim 1 \times 10^{12} \text{ cm } \sqrt{\text{Hz}} \text{ W}^{-1}$.

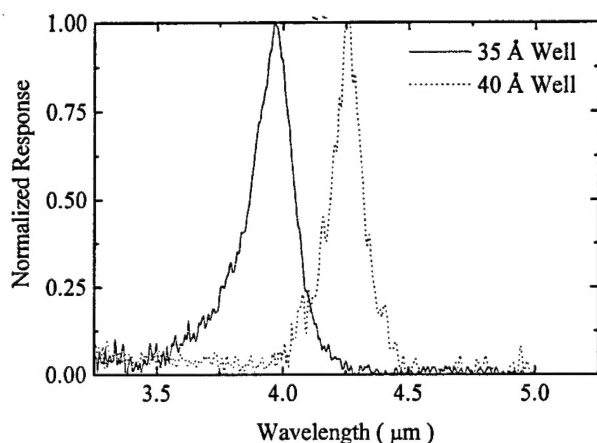


Fig. 4. Spectral response for $\text{In}_{0.53}\text{Ga}_{0.47}\text{As}/\text{In}_{0.52}\text{Al}_{0.48}\text{As}$ QWIPs for well widths of 35 and 40 Å.

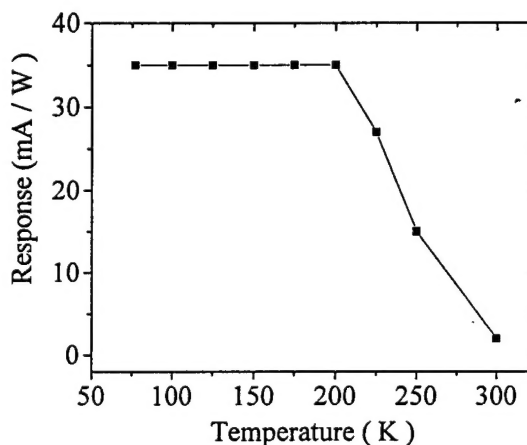


Fig. 5. Temperature dependence of peak responsivity for the 40 Å $\text{InGaAs}/\text{InAlAs}$ QWIP sample.

V. Dual Band Infrared Detector

A sample was grown for multispectral absorption in both MWIR and LWIR regions by including multiple quantum wells of both $\text{InGaAs}/\text{InAlAs}$ and InGaAs/InP . Because the mask utilized in this work allows only one contact to be made to the mesa, the two MQW detectors are placed in series. At low biases, the electric field is applied mostly across the lower-resistance InGaAs/InP MQW. At higher biases, the electric field is applied across both MQWs, but the InGaAs/InP MQW is under such high field that no response is observed. Most carriers in the ground state tunnel out before absorption occurs, resulting in low LWIR photoresponse. A schematic of the processed device is shown in Fig. 6. The spectral response of this dual-band detector is shown in Fig. 7. for several biases. For biases less than 10V, the photoresponse in the 3-5 μm region is too noisy to resolve. For biases greater than ~ 7 volts, the photoresponse in the 8-9 μm region is no longer observed.

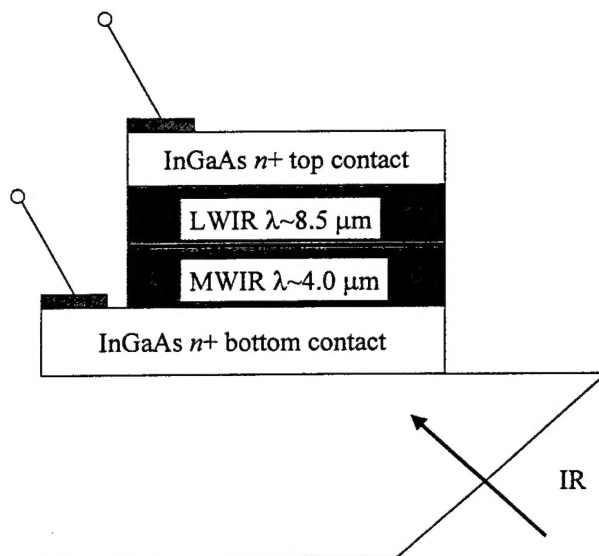


Fig. 6. Schematic diagram of a processed two-color detector.

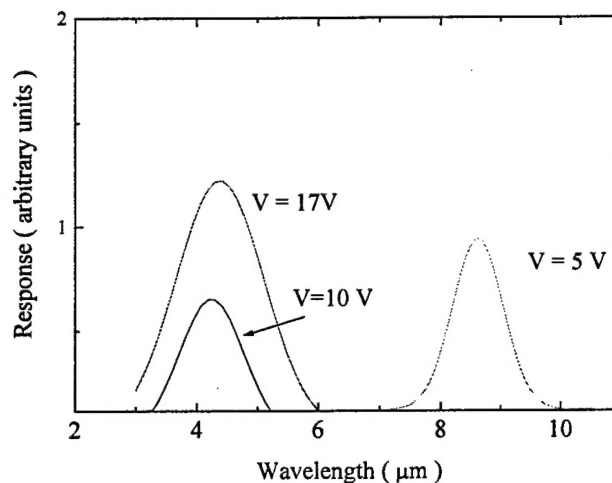


Fig. 7. Voltage dependence of the spectral response for the two-color detector, measured at $T=77\text{K}$.

GaAs-Based QWIP Results

I. Structure.

GaAs-GaInP n-type QWIP structures were grown using gas-source molecular beam epitaxy (GSMBE) on (100) semi-insulating GaAs substrates. GaAs well widths were 40 Å, 65 Å, and 75 Å respectively, and the lattice-matched $\text{Ga}_{0.51}\text{In}_{0.49}\text{P}$ barriers were 500 Å thick. Each sample superlattice had 20 periods. The silicon doping in the quantum wells was $5 \times 10^{17} \text{ cm}^{-3}$.

II. Responsivity.

The photoresponse for three different well widths is shown in Fig. 8. The peak wavelengths of the three samples are 10.4, 12.78, and 13.3 μm for well widths of 40, 67, and 75 Å respectively. The cutoff wavelengths are 13.5 μm, 15, and 15.5 μm for the three samples. The full width at half maximum (FWHM) for the three samples in meV (and as $\Delta\lambda/\lambda$) were 185 meV (0.58), 37 meV (0.5), and 42 meV (0.375) respectively. These FWHM are typical for bound to continuum QWIPs. The reduction in FWHM for wider wells indicates that the second excited state in these samples is closer to resonance with the barrier energy.

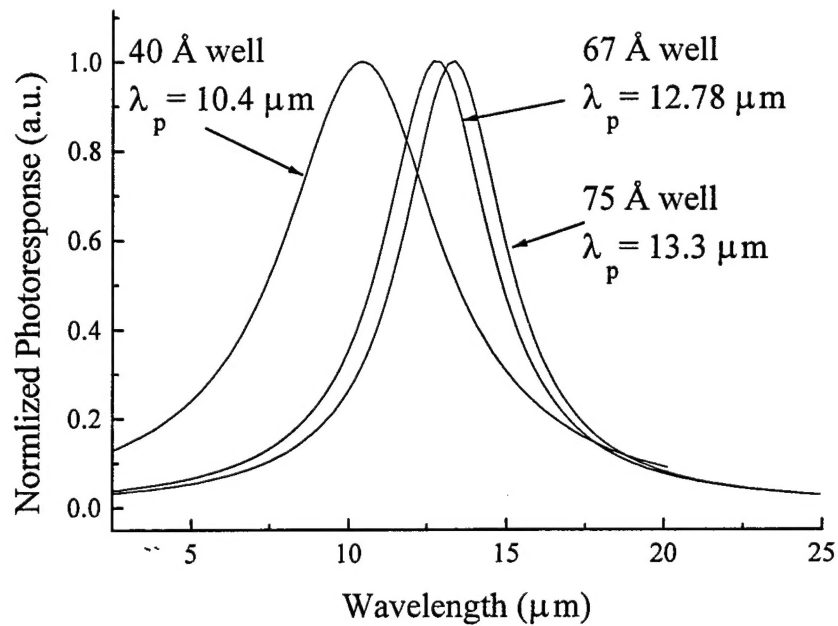


Fig. 8. Normalized optical response at 77K for samples with 40Å, 65Å and 75Å quantum wells.

Shown in Fig. 9 is the calculated absorption coefficient for a 75Å well GaAs/GaInP QWIP. The calculated peak absorption of 12.9 μm and an approximately 14 μm cutoff wavelength agree very well with the observed photoresponse shape.

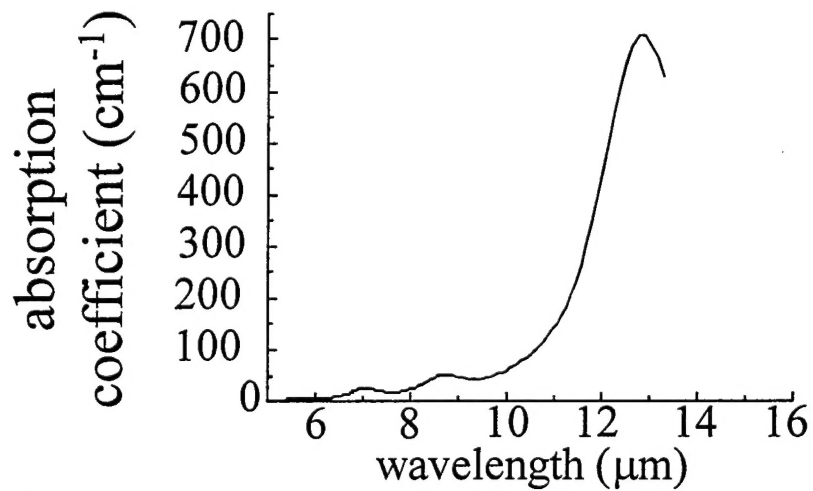


Fig. 9. Calculated absorption spectrum for 75Å GaAs / 500Å GaInP superlattice.

III. Dark Current.

Figure 10 shows the typical dark currents measured at different temperatures. The observed asymmetry with bias may be due to an asymmetric quantum well profile or doping impurity segregation.

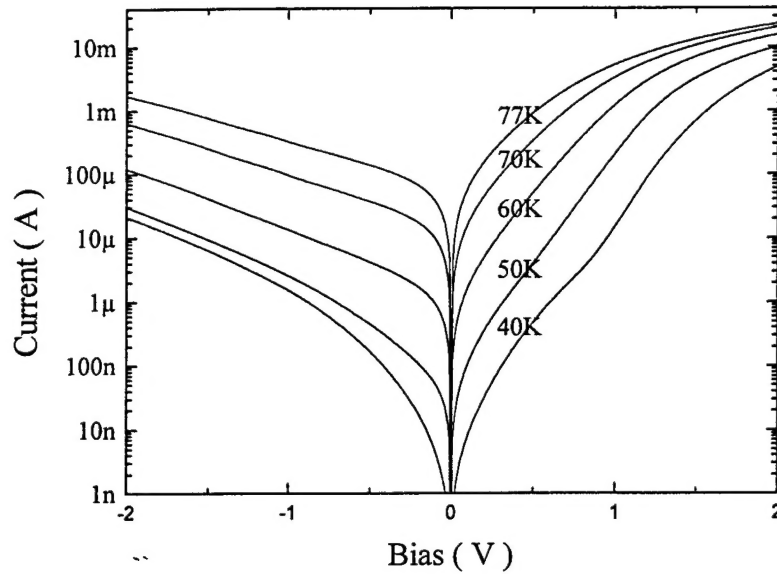


Fig. 10. Measured QWIP dark current from 30 to 77 K for the sample with 65 Å quantum wells.

At low bias, where tunneling can be neglected, the dark current I is expected to increase exponentially with temperature following equation (1):

$$I \propto T \exp\left(-\frac{\Delta E}{kT}\right) \quad \{1\}$$

Therefore, the activation energy can be obtained by calculating the slope of $\log(I/T)$ versus $1/kT$. For the three samples with 40Å, 65Å, and 75Å wells, the activation energies ΔE are 37, 50, and 46 meV respectively.

The quantum well subband energies were determined using a four-band Kane model, which includes the effects of band non-parabolicity and band mixing. The electron-electron exchange interaction effect was also included, and the depolarization and excitation shifts were neglected as before.

From these subband energy levels and the dark current activation energies we computed the conduction band offset ΔE_c to be $121.1 \text{ meV} \pm 2 \text{ meV}$. A summary of the data used in this model is given in Table I. From the known band gaps of GaAs and GaInP, the band-gap difference $\Delta E_g = 483 \text{ meV}$ is obtained¹⁸. Using our experimentally determined conduction band offset, a band ratio offset ratio of $\Delta E_c / \Delta E_g = 0.251$ is calculated, which matches the assumed ratio used for theoretical modelling. Values of ΔE_c in the literature range between 80¹⁹ to 240⁵ meV. However, the only previous measurement for material grown by GSMBE was 108 meV²⁰. This experimental result agrees well with these previous results.

TABLE I.

THE CONDUCTION BAND OFFSET IS CALCULATED AS $\Delta E_c = (E_o - E_{EXCH}) + E_F + \Delta E$, WHERE E_o IS THE GROUND STATE, E_{EXCH} IS THE ELECTRON-ELECTRON INTERACTION ENERGY, E_F IS THE FERMI ENERGY, AND ΔE IS THE ACTIVATION ENERGY.

Well (Å)	$E_o - E_{EXCH}$	ΔE	E_F	ΔE_c
40	79	37	4.4	120.4
65	62	50	10.4	122.4
75	59	46	12.5	120.5

IV. Modeling

A. DARK CURRENT

1. Comparison to GaAs/AlGaAs:

The dark current density (A/cm^2) at 77K for the 65Å sample as a function of electric field (V/cm) is shown in Figure 11. The results of Levine [1] for a GaAs/AlGaAs QWIP with similar well doping density, and cutoff wavelength is shown for comparison. The smaller dark current may be due to GaAs/ $\text{Ga}_x\text{In}_{1-x}\text{P}$'s larger mobility [9], smaller surface recombination velocity [10], or smaller interface roughness [11] by comparison to GaAs/ $\text{Al}_x\text{Ga}_{1-x}\text{As}$.

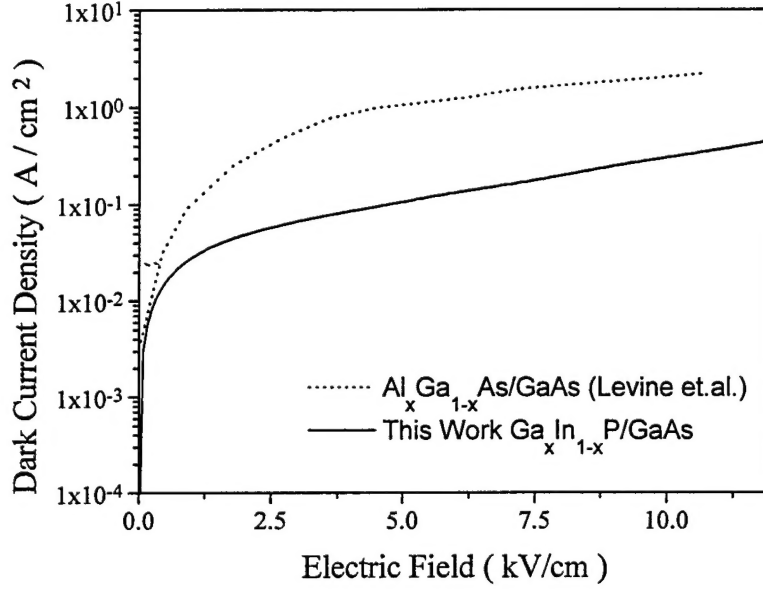


Figure 11 Measured dark current density for sample C compared to GaAs/AlGaAs results in reference [Error! Bookmark not defined.].

2) Modeling:

We have modeled the dark current by assuming thermodynamic carrier equilibrium and a drift model [12] where the drift velocity is proportional to the electric field according to:

$$v_d = \frac{\mu V}{L_b} \left[1 + \left(\frac{\mu V}{v_s L_p} \right)^2 \right]^{-1/2} \quad \{2\}$$

where μ is the mobility, V is the applied bias per quantum well, and L_p is the period thickness. We have neglected any complications to do with the way in which carriers in the quantum well are replenished and the mechanisms of charge transfer from the emitter contact to the MQW. Since the carriers which are thermally excited into the continuum transport states and contribute to the dark current are originally from the quantum wells, a 2-D density of states (DOS) is used. The effective number of carriers contributing to the current is then:

$$n = \left(\frac{m^*}{\pi \cdot \hbar^2} \cdot \frac{1}{L_p} \right) \int_{E_o}^{\infty} f_o(E) T(E, V) dE, \quad \{3\}$$

where the term $m^*/\pi \hbar^2$ is the 2-D DOS, $f_o(E, T)$ is the Fermi-Dirac distribution function, $T(E, V)$ is the probability of tunneling through a single barrier, and E_o is the bound state energy (measured from the bottom of the band). In

this equation, the reason why the 2-D DOS is divided by the period width is to match the units, i.e. to convert sheet concentration into an average volumetric concentration. The dark current density is calculated using equation:

$$J = nev_d \quad \{4\}$$

In calculating the transmission coefficient $T(E,V)$ we have used $T(E,V)=1$ for E larger than the conduction band offset V_o and the Wentzel-Kramers-Brillouin (WKB) approximation [13] for $E < V_o$. In the calculation we take the conduction band offset V_o to be 0.1 eV. Conduction band non-parabolicity is neglected because it has been shown to have a negligible effect on dark current [14], but image charge effects have been included.

The dark current measurements for the 65Å sample are shown in Fig. 12 as function of voltage and temperature and compared with those calculated by equation {4}. Good agreement is achieved as a function of both bias and temperature over seven orders of magnitude in dark current. These calculations use a drift mobility of $1000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$, a saturation velocity of $1.5 \times 10^5 \text{ cm}^{-1}$ and the nominal sample parameters shown in Table I. The saturation velocity is approximately one order of magnitude lower than typical values for bulk GaInP at 77K [15,16] in a similar electric field, as would be expected because of re-trapping by quantum wells and interface scattering.

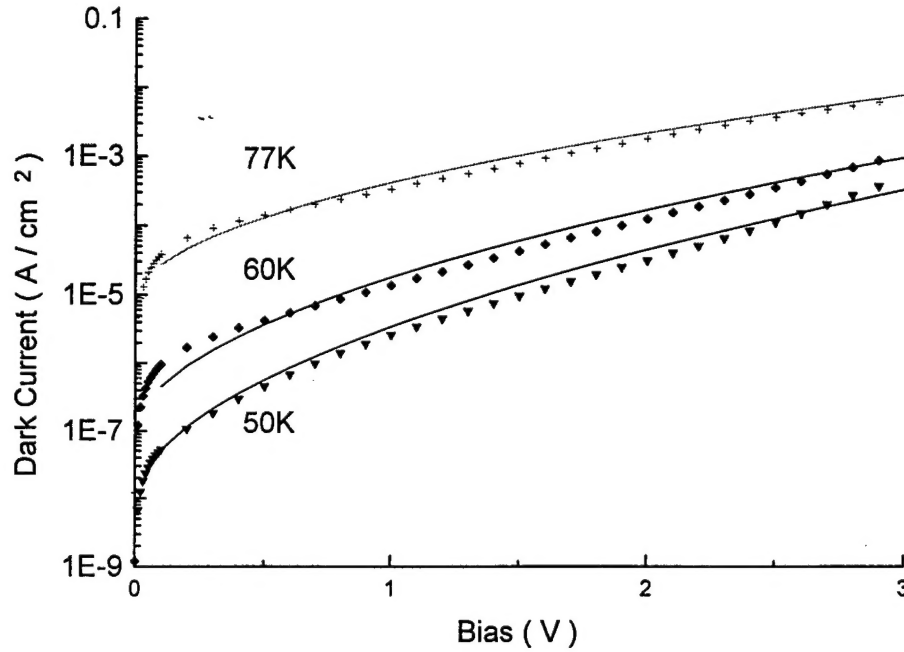


Figure 12 Comparison of experimental (points) and theoretical (lines) dark current-voltage curves at various temperatures for sample B.

The photoconductive gain is ratio of the distance traveled by an electron before recapture l to the thickness of the device L . At the biases used in this experiment, $l = v_d(E) \cdot \tau$ and therefore

$$g = v_d(E) \cdot \tau / L. \quad \{4\}$$

In order to evaluate the carrier lifetimes in these GaAs/GaInP detectors, we have directly measured the noise current as a function of bias using a spectrum analyzer at $T=77\text{K}$. This noise is dominated by generation-recombination noise at $T=77\text{K}$. From the equation for G-R noise given by $I_N = (4e \cdot I_d \cdot g)^{0.5}$, we calculate the gain using the dark current-bias data from Fig. 10, and then calculate the carrier lifetimes using equation {4}. Figure 13 shows the carrier lifetimes and gain derived from noise measurements for the 65Å sample.

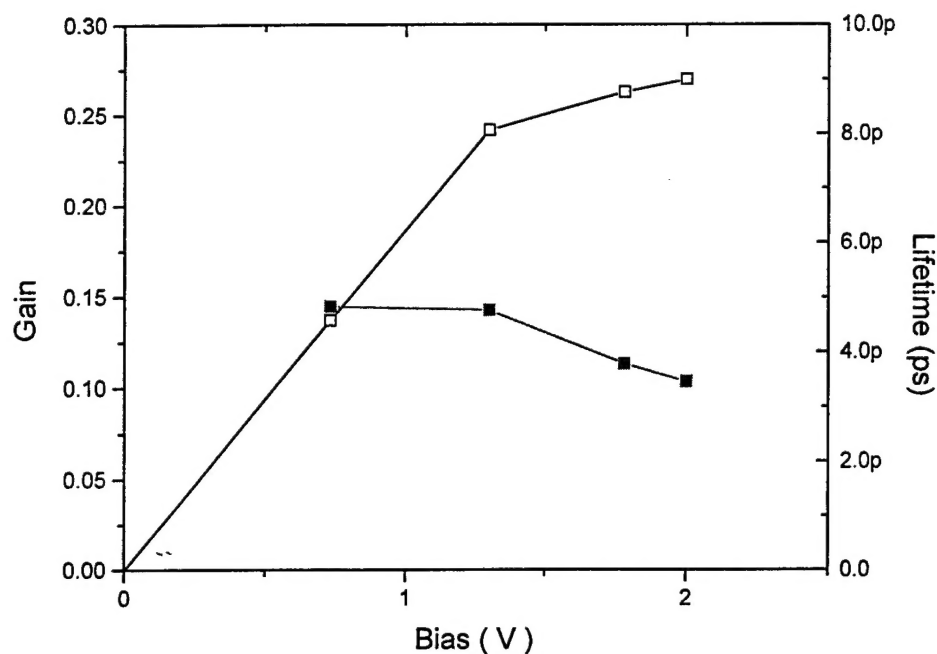


Figure 13. Calculated gain and recombination lifetimes as a function of applied bias for sample B.

Conclusion

We have demonstrated the first QWIP detectors using the quaternary InGaAlAs/InP materials system. We have observed excellent responsivity in the InGaAlAs/InP system compared to GaAs/AlGaAs QWIPs. By increasing the bandgap from ternary InGaAs to quaternary InGaAlAs we have shifted the responsivity out to longer wavelengths resulting in cutoff wavelengths of 13.3 and 19.4 μm s, respectively for AlAs mole fractions of 0.1 and 0.15. We have also demonstrated lattice-matched mid-wavelength infrared detectors using InGaAs/InAlAs quantum wells. By combining both types of devices, we have produced the first lattice-matched dual band mid-wavelength and long-wavelength QWIP detectors on InP substrate. We have also demonstrated long-wavelength QWIPs ($\lambda > 12 \mu\text{m}$) in GaAs-GaInP structures, and theoretically calculated the performance of these devices.

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